# Comparative Study on the Tribological Properties of AISI52100 Steel Surface Modified by Graphite/MoS<sub>2</sub> Composite Coating and Manganese Phosphate Coating

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Surface modification is an important method to improve the contact fatigue life of transmission parts. The preparation of high performance coating with good tribological properties on the gears and bearings has become the research trends. This paper presents graphite/MoS<sub>2</sub> composite spray and manganese phosphate conversion coating prepared on the AISI52100 steel surface and investigates their anti-fatigue mechanism. The tribological properties of the modified layers were studied using a SRV-IV multifunctional friction and wear tester. The microstructure and interfacial components of the coating and wear surfaces were analyzed by SEM and EDS, respectively. The surface morphology and phase composition of the coating were evaluated through laser 3D microscopy and XRD analyses, respectively. Both modified layers showed good anti-friction and anti-wear properties. The friction coefficients of the surfaces modified by manganese phosphate coating and graphite/MoS<sub>2</sub> decreased by 7 % and 14 %, respectively, and the corresponding extreme pressure properties increased by 11 % and 55 %, respectively. But the mechanism of anti-fatigue wear and the corresponding interfacial phenomena of the surface modification of the graphite/MoS2 composite spraying layer is different from those of manganese phosphate coating. The hard-ceramic particles with graphite and MoS2 are sprayed on the substrate to obtain the surface hardening layer, resulting in higher wear resistance. The graphite and MoS2 modified layer can greatly reduce the friction coefficient and improve the lubrication performance on the surface. Manganese phosphate coating serves as a chemical soft coating and is filled with rough corrugated caused by surface processing, which induces an ideal meshing surface after initial friction phase. The obtained hole-shaped structure and "infiltration" role contribute to the storage of lubricants and thus improves the lubrication performance.

Keywords: surface modification, spraying, electrochemical deposition, tribology.

## **1. INTRODUCTION**

Gear drive is the most important form of mechanical transmission [1]. In this process, gears are subjected to various forces, including bending stress, contact stress, and impact force, which lead to formation of failures, such as tooth flank pitting, spalling, wear, scoring, and tooth bending fatigue [2, 3]. Among these failures, fatigue pitting and scuffing are typical gear damage patterns, which affect transmission and the lifetime and reliability of other driveline systems [4]. Friction and wear occur in mechanical components with contact interfaces. During high-speed rotation movement, high contact stress and meshing speed under repeat forces lead to a certain extent of friction on the gear surface up to the matrix, resulting in gear surface fatigue pitting and scuffing. To solve these limitations, scholars primarily use surface modification for improving the gear anti-fatigue damage capability.

Studies on gear anti-fatigue surface modification have focused on surface coating and shot peening [5-13]. Shot peening is a new technology that is categorized into: laser shock peening [8], ultrasonic shot peening [9], fine particle peening [10], and water cavitation peening [11]. Gear surface coating modification commonly utilizes TiN coating [12], MoS<sub>2</sub>/Ti membrane [13], and phosphate coating. Lv Y. et al. investigated the effect of shot peening modification and radar surface processing parameters on the fatigue property of 20CrMnTi gear; results showed that the fatigue lifetimes of the LSMSSP gears are grain determined by the retained austenite and surface roughness [5]. Shi et al. studied the influence of manganese phosphate coating on the heat scuffing property of a worm and gear system and found that the manganese phosphate coating can improve the anti-scuffing ability of worm gearing [14]. With the development of transmission systems featuring high strength, high speed, long lifetime, light weight, and downsizing directions, traditional singlesurface modification becomes unsuitable and insufficient [4]. Shot peening is an effective method used to improve the anti-bending fatigue property of gear surface but affects the contact fatigue damage on the gear surface. This study aims to investigate the mechanism of the high contact fatigue lifetime of gears and bears with graphite/MoS<sub>2</sub> spray and chemically deposited manganese phosphate coating. Graphite/MoS<sub>2</sub> spray and manganese phosphate coating were prepared on a standard low-carbon alloy steel. The test rig was comprehensively evaluated and compared. The interface microstructure and friction property of the modified layers were analyzed by SEM, XRD, and tribological properties test. The interface

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mechanism sunder lying the reduction of anti-fatigue and friction were revealed.

## 2. MATERIALS AND EXPERIMENTAL

## 2.1. Substrate

The material components of the test specimens are shown in Table 1, the material is a standard specimen provided by OPTIMOL, and basic information, such as size and surface treatment of the specimens, are illustrated in Table 2. The substrate is a bearing steel (AISI52100, American Standard), which was smelted by vacuum arc D method. The bearing steel is classified as Type A, has a severity level number of 0, 5 (according to Test Methods E45 and Specification A295), and an inclusion sum value of  $K1 \le 10$  (according to DIN EN ISO 683-17). The bearing steel was spheroidized and annealed to obtain globular carbide ( $62 \pm 1$  HRC hardness). The surfaces of the disk were lapped and had no lapping raw materials. The topography of the disk was determined by four values:  $0.5~\mu m < Rz < 0.650~\mu m,~0.035~\mu m < Ra < 0.050~\mu m,$ 0.020  $\mu m < Rpk < 0.035$   $\mu m,$  and 0.050  $\mu m < Rvk < 0.075$ um.

Table 1. Composition of the tested material, wt.%

С	Mn	S	Р	Cr	Mo	Ni	Cu	Fe
0.1	0.18	0.02	0.017	1.5	0.08	0.25	0.02	Balance

**Table 2.** Specimen materials and their treatment conditions

Specimen type	Specimen size	Material	Treatment	Hardness, HRC
disc	Ø 24 × 7.9 mm	GCr15	Quenching (820 ~ 860 °C, oil quenching) Tempering ( $60 \pm 10$ °C, 2 ~ 4 h)	61 ~ 63

## 2.2. Sample preparation

## 2.2.1. Graphite/MoS<sub>2</sub> modification

A graphite/MoS<sub>2</sub> (PC/MoS<sub>2</sub>) layer was prepared as follows. The first type of PC/MoS<sub>2</sub> was prepared mixing the hard- ceramic beads with the solid lubricant in the ejector device. This study selected hard ceramic rolling balls with diameter of 50 µm whose main components of hard ceramic balls are ZrO2 and SiO2, where  $60 \ \% < ZrO_2 < 70 \ \%$ ,  $28 \ \% < SiO_2 < 33 \ \%$ .  $MoS_2$  and graphite mixture were used as solid lubrication material, and the mass ratio of ceramic balls and solid lubrication is 20:1. The second type of PC/MoS2 was prepared heating the mixture in the ejector to 300 °C, using inert (N<sub>2</sub>, 0 °C) gas under the pressure of 0.4 MPa to push hard particles and solid lubricant onto the surface of the specimen. Hereafter, the surface modification layer sample of PC/MoS<sub>2</sub> was obtained to alter the phase structure of the metal surface at 20 µm in depth [4]. Finally, the ceramic balls were recycled by the recycling device.

## 2.2.2. Manganese phosphate coating

Electrochemical deposition was employed to prepare manganese phosphate coating (Mn–P coating). Briefly, the

steel specimens were cleaned, the substrates were firstly subjected to degreasing treatment at 70 °C ~ 95 °C for 5 min. The degreasing agent was a kind of weak alkaline degreasing agent, whose composition was Na<sub>3</sub>PO<sub>4</sub>, Na<sub>2</sub>CO<sub>3</sub>, etc. These samples were separately sonicated in acetone, ethanol and deionized water for 15 min, and dried in N<sub>2</sub> atmosphere. After water cleaning, the surface was modified at 40 to 80 °C and underwent phosphating treatment at 80 ~ 100 °C for 10 to 15 min while controlling the acid ratio between 5.6 and 6.2. The phosphating solution mainly contained of Mn<sup>2+</sup>  $\sim$  Fe<sup>2+</sup>, Ni<sup>2+</sup>, PO<sub>4</sub><sup>3-</sup>, NO<sub>3</sub> etc.

#### **2.3.** Characteristics

#### 2.3.1. Microstructure characteristics

Morphology and composition were investigated through scanning electron microscopy (SEM, JSM-7100F, Japan) and energy dispersive spectroscopy analyses (EDS). Surface roughness was calculated using a 3D measurement laser microscope (CLSM, OLYMPUS LEXT OLS4000, Japan). The phase composition of the modified layer was investigated by X-ray diffraction (XRD, D8 Focus Bruker AXS, Germany).

#### 2.3.2. Friction coefficient experiment

The tribological properties of  $PC/MoS_2$  and Mn-P coatings were tested on an SRV multifunctional friction test machine (SRV-IV, OPTIMOL, Germany). The equipment and test principle are shown in Fig. 1 The SRV friction test equipment can simulate movement, displacement, and direction of friction contact, including spot contact, linear contact, and face contact; the movement pattern can simulate reciprocal movement and rotation movement and can be applied in lubrication, material, and coating friction analysis. This work used spot contact and the reciprocal movement pattern to measure the sliding friction coefficient of the coatings.

Multi-pass reciprocating sliding tests were carried out in a translator oscillation tribometer (Fig. 2 b), which simulates well the reciprocating movement. The experiment standards according to ASTM standards D7421 and experiment parameters are shown in Table 3. Each test was performed under the same conditions for three times for statistical evaluation. New surface was selected for each test. The property parameters of lubrication oil are shown in Table 4. Before the experiment, to make the surface clean and avoid surface impurities affecting the accuracy of measurement of friction coefficients, all the samples were subjected to ultrasonic cleaning for 15 min with n-heptane before each test and therefore, the tests were done under lubricating of adopts disposable titration 0.3mL of lubrication oil.

Table 3. The friction and wear test parameters

Parameters	Value
Temperature, °C	80
Frequency, Hz	50
Stroke sliding speed average value current, m/s	0.1
Load, N	50/200
Stroke, mm	1
Time, s	6000



Fig. 1. SRV measurement equipment and measurement principle: a-SRV measurement equipment; b-measurement principle diagram

**Table 4.** Combination of the oil specifications of an experiment

Factor	DCTF
Density, g⋅cm <sup>-3</sup> @15 °C	0.867
Dynamic viscosity, cSt @100 °C	7.55
Mean dynamic friction coefficient	0.128
Lubricating oil volume	0.3 mL

In tribology, the coefficient of friction (f), n is the dimensionless ratio of the friction force  $(F_f)$  between two bodies to the normal force  $(F_n)$  pressing these bodies together.

$$f = \frac{F_f}{F_n}.$$
(1)

## 2.3.3. Extreme pressure (EP)

The SRV test equipment was used to measure two kinds of surface modification EP properties to evaluate the effect of surface modification methods on the EP property of lubrication oil. In the test procedure according to ASTM D4172, a dynamic test program with the following test parameters was generated. In this dynamic test program, the maximum load range should be set to a maximum of 2000 N with a frequency of 50 Hz for a stroke length of 50 mm. The test temperature of 80 °C and a load of 50 N was applied for 30 s, followed by a load increment of 100 N for 15 min. A load increment of 100 N was then applied every 2 min.

Load carrying capacity can be reported as geometric contact pressure  $P_{\text{geom}}$  at test end according to the following equation:

$$\mathbf{P}_{\text{geom}} = \frac{F_N}{\pi W_K^2},\tag{2}$$

where  $F_N$  is normal force (test load);  $P_{geom}$  is geometric contact pressure, and  $W_k$  is the wear scar diameter.

#### 2.3.4. Observation of morphology after friction

After the friction test, the test parts were ultrasonically cleaned. The surface morphology and element composition of the friction test parts were studied through SEM and EDS.

## **3. RESULTS AND DISCUSSION**

In the present study, Fig. 2 a and b show the results of applying hard micro particles, graphite, and  $MoS_2$  solid lubrication mixture to the coating on the bearing outer ring; this method reduced the pressure marks in the transmission needle bearing and the surface friction [4].





Fig. 2. Examples of spray and manganese phosphate coating modification: a-nedle-bearing outer ring prior to treatment macroscopic morphology; b-graphite/MoS<sub>2</sub> after treatment macroscopic morphology; c-gear surface before treatment through testing for  $1.2 \times 10^6$  cycles; d-manganese phosphate after treatment through  $1.2 \times 10^7$  cycles

Fig. 2 c and d show the prepared manganese phosphate coating after carburizing and quenching heat treatment. The fatigue experiment results indicated that gear contact fatigue lifetime was enhanced by two to three times after

treatment of the manganese phosphate coating [15, 16]. However, the principles of the two surface treatments and interface analysis remain unclear.

## 3.1. Surface microstructure

#### 3.1.1. Macroscopic morphology

The matrix samples were treated by two surface modification methods using PC/MoS<sub>2</sub> and Mn–P coating to obtain the surface modification layer. Fig. 3 shows the macroscopic morphologies of manganese phosphate coatings. Evidently, the PC/MoS<sub>2</sub> and Mn–P coatings completely mask the metallic luster of the matrix, and the surface of the manganese phosphate coating is more uniform and denser than that of the PC/MoS<sub>2</sub> surface.





Fig. 3. Specimens before and after treatment; a-untreated; b-modified with PC/MoS<sub>2</sub>; c-modified with P-Mn coating

#### 3.1.2. Structural examinations

The phase structure on the surface of the coating specimen was analyzed with X-ray diffraction (XRD). Fig. 4 a shows that the main phase of the film obtained after manganese phosphate conversion treatment is manganese phosphate  $Mn_3(PO_4)_2 \cdot 3H_2O$ . Small amounts of  $Mn_2P_4O_{12} \cdot 10H_2O$  and (Fe, Mn)<sub>2</sub>(PO<sub>4</sub>)(OH), and iron oxide were formed on the contact surface between the film and the substrate. The reaction principle of manganese phosphate coating is as follows:

 $Mn(H_2PO_4)_2 \rightleftharpoons MnHPO_4 + H_3PO_4$ 

 $Fe(H_2PO_4)_2 \rightleftharpoons FeHPO_4 + H_3PO_4$ 

 $3(Mn, Fe) (H_2PO_4)_2 \rightleftharpoons (Mn, Fe)_3(PO_4)_2 + 4H_3PO_4$ 

As shown in Fig. 4 b, the  $PC/MoS_2$ -modified surface mainly consists of solid lubricants, such as  $MoS_2$  and

graphite. The XRD patterns show the successful preparation of two modified layers on the substrate surface.



**Fig. 4.** Surface XRD pattern of Mn–P coating and PC/MoS<sub>2</sub>: a-phase structure of the Mn–P coating; b-phase structure of PC/MoS<sub>2</sub>

## 3.1.3. Surface morphology

Fig. 5 shows the SEM image (JSM-7100F) indicating the surface morphologies of the surface modified by PC/MoS<sub>2</sub> and Mn-P coatings. The PC/MoS<sub>2</sub>-modified layer form sadense solid lubricant and binder, and the coating surface is smooth (Fig. 5 a). The formation of manganese phosphate conversion coating is a chemical conversion process between the steel surface and phosphate solution, and the coating is formed after the reaction between the solution and the substrate [4]. The coating is irregularly arranged with a small amount of voids on the surface. The size of the phosphating grain can be regulated by controlling the reaction process. The grain size of the phosphating film prepared in this experiment is 5-7 µm. Based on the equation  $Fe+2H^+ \rightleftharpoons Fe^{2+}+H_2$ , iron is oxidized by hydrogen ions in acidic solutions, and the neutralization reaction leads to the deposition of small amounts of soluble phosphate. During the reaction, a mixed phosphate is formed due to the substitution of Fe<sup>2+</sup> with Mn<sup>2+</sup>.The deposited oxide layer is discontinuous, as verified by the surface micromorphology of the coating.

## 3.1.4. Surface roughness

The tribological properties of the contact surfaces are directly affected by the surface morphology.



**Fig. 5.** Surface topography: a – PC/MoS<sub>2</sub>, b – Mn – P coating

h

10um

Roughness is the most commonly used surface topography parameter that describes the geometrical characteristics of a surface [18]. The effect of the initial friction surface topography on the characteristics of wear performance occurs in two aspects: surface morphology affects the working performance of friction pairs in steady wear stage after running and wear rate affects the working life of friction pairs during the running-in process. Surface roughness significantly influences the lubrication conditions of gears and other contact pairs. The rough surface can reduce the lubricant film thickness between the gears and deteriorate the lubrication conditions. The surface topography of PC/MoS<sub>2</sub> and Mn-P coatings were analyzed using a laser microscope (Fig. 6). The surface roughness of the coatings was then measured three times, and the average value was determined. The line roughness and surface roughness of the PC/MoS<sub>2</sub> surface are  $Ra = 0.101 \ \mu m$  and  $Sa = 0.329 \ \mu m$ , respectively, whereas those of the Mn-P coating are  $Ra = 0.515 \,\mu m$  and  $Sa = 0.501 \mu m$ , respectively. Compared with that of the substrate, the surface roughness after surface modification is slightly improved due to the formation of iron oxides (XRD results). However, the two surface-modified coatings belong to the soft layer, which can reduce the surface roughness after the initial run-in. This finding can be proven by the following friction test.

## **3.2.** Microstructure of section

Fig. 7 shows the surface topography and energy spectra of the  $PC/MoS_2$  surface-modified specimen. The

lubricant layer thickness is  $8-12 \mu m$ , and the coating is loosely organized from the cross section (Fig. 7 a). Fig. 7 b, c, and d show the EDS maps of the following elements on the surface: 1 % Mo and 11 % C. Mo, S, and C are present on the interface between the coating and the substrate.

Fig. 8 shows the cross-section topography and distribution of element of the Mn – P coating. As shown in Fig. 8 a, the thickness of the lubricant layer is  $5 \sim 7 \mu m$ . Based on analysis of the cross section, the coating is compact in structure and bonds well to the substrate. The element matrices of Mn and P diffuse at the interface of the coating and matrix, and the contents of P and Mn are 11 % and 6 %, respectively (Fig. 8 b, c, and d).

## 3.3. Tribological properties

#### 3.3.1. Friction coefficient

Friction caused by the sliding movement of two gear surfaces leads to fatigue pitting friction and adhesion friction. The sliding friction coefficient is the key tribological property of the surface modification layer [17]. Sliding friction coefficient can be measured to evaluate the tribological performance of the two kinds of surface layers. According to the friction coefficient—time curve, the status and mechanism of the modified surface during friction were further analyzed.

Fig. 9 shows the sliding friction coefficient test curves of the two surface-modified layers, as measured by the SRV friction tester. During the period with a load of 50 N, the specimen friction coefficient decreased after treatment with PC/MoS<sub>2</sub> and Mn-P coating, and the average sliding friction coefficient of the untreated substrate is between 0.150 and 0.155 (Fig. 9). The friction coefficient decreased to 0.05 to 0.1 after the two surface modification processes, and the friction coefficients of PC/MoS2 and Mn-P coating are basically the same. In the 200 N load grinding stage, the friction coefficient curve decreased, and the friction coefficient of the Mn-P coating decreased by 0.07-0.09, or about 7 %, compared with that of the untreated sample. PC/MoS<sub>2</sub> exhibited improved friction reduction performance, and the friction coefficient decreased by approximately 0.2, showing 14 % reduction.

#### 3.3.2. Extreme pressure characteristics

Good extreme pressure property affects gear lifetime. Extreme pressure determines the loading capability of transmission components and lubrication conditions under large alternating loading forces [19]. Extreme pressure property test is used to investigate gear lubrication oil loading and anti-friction capability.

Fig. 10 shows the extreme pressure characteristic curve of the untreated substrate,  $PC/MoS_2$ , and Mn-P coating. The extreme pressure load of the untreated specimen is 900 N, and the friction curve is relatively smooth (Fig. 11 a). The extreme pressure load of the Mn-P coating is1000N, the friction curve jumps with a load of 600 N (Fig. 10 b). The phosphate coating possibly formed flakes, with the low friction coefficient and good self-lubricating properties make the two coatings have better extreme pressure characteristics.



**Fig. 6.** Three-dimensional topography observation and roughness measurement: a – CLSM picture of PC/MoS<sub>2</sub>; b – CLSM picture of the Mn–P coating; c – CLSM three-dimensional topography of PC/MoS<sub>2</sub>; d – CLSM topography of the Mn–P coating



**Fig. 7.** Cross-section morphology and EDS of PC/MoS<sub>2</sub>: a-cross-sectional morphology; b-elemental mapping distribution; c-mapping distribution of Mo element; d-mapping distribution of C element



Fig. 8. Cross-section morphology and EDS of Mn-P coating: a – cross-sectional morphology; b – elemental mapping distribution; c – mapping distribution of Mo element; d – mapping distribution of C element



Fig. 9. Friction coefficients of untreated specimens,  $PC/MoS_2$ , and Mn - P coating

Fig. 10 c shows that  $PC/MoS_2$  bears a high extreme pressure load of 1400 N.Hence,  $PC/MoS_2$  improves the extreme pressure performance of lubricating oils, and the modified surface exhibits good abrasion resistance.

## 3.4. Observation of morphology after friction

### 3.4.1. SEM

Fig. 11 shows the SEM image indicating the surface topography of  $PC/MoS_2$  and Mn-P coating specimens after the friction test. After wear test, the  $PC/MoS_2$  surface lubrication bonding coating is damaged and some of abrasive marks. There's a partial spalling of the coating.

(Fig. 11 a). However, the base material of the specimen is protected by the lubrication layer and there is no obvious wear mark. It is assumed that the adhesive wear does not happen on the substrate surface. But the lubricants oil color came black, it is inferred that graphite and  $MoS_2$  solid coatings dissolved into lubricating oils. Moreover, the original trace of friction pairs is clearly visible on the surface, showing markedly improved abrasion resistance of the matrix.

Given that the manganese phosphate modified coating is classified as a chemical combination, the coating shows good bonding property, the surface does not show the peeling phenomenon, and abrasion occurs in the coating. As can be seen from Fig. 11 b, the sliding at a contact load of 200 N after 6000 s produced a smooth wear track with the phosphate coating still present on the substrate.

#### 3.4.2. EDS analysis

Fig. 12 shows the results of the EDS element analysis of the wear surface after wear testing on the two kinds of surface-modified specimens. Fig. 12 a and c show that the S and Mo contents in the PC/MoS<sub>2</sub> wear surface account for 2 %, and the surface of the coating remains to be lubricated by the self-lubricating material. Fig. 12 b and d show the wear marks of Mn–P, and the coating wear surfaces possess high amounts of Mn and P.

#### **3.5. Friction mechanism analysis**

Fig. 12 a and c show the element content of a point (EDS) on the  $PC/MoS_2$  coating.



Fig. 10. Extreme pressure characteristics of untreated specimens, Mn-P coating and  $PC/MoS_2$ : a-untreated specimens; b-Mn-P coating;  $c-PC/MoS_2$ 



Fig. 11. SEM morphology of PC/MoS<sub>2</sub> and Mn-P coatingafter the friction test: a-PC/MoS<sub>2</sub>; b-Mn-P coating

The wear surface contains high amounts of Mo, S, and C. Mo at the interface between the coating and substrate diffuses into the matrix, and a large amount of Mo, S, C, and other elements are enriched at the parent material near the interface. This elemental distribution not only strengthens the matrix but also play the role of antifriction solid lubricant and extreme pressure additive. As seen from the wear morphology, the furrow effect and adhesion wear on the surface become abrasive wear owing to MoS<sub>2</sub>,

graphite particles fall off, and the transfer of film lubrication occurs. The possible antifriction and antiwear mechanisms are as follows: the solid lubricant, such as  $MoS_2$  and graphite, enters into the friction pair surface, remains in the rough trough or grinding mark of the friction pair surface, and increases the actual contact area of the friction pair surface; the abrasion-generated grinding marks are filled at any time by the  $MoS_2$  solid lubricant.



Fig. 12. Analysis of surface energy spectrum components of PC/MoS<sub>2</sub> and Mn–P coating: a–PC/MoS<sub>2</sub> grinding EDS element distribution; b–PC/MoS<sub>2</sub> element content; c–Mn–P coating grinding EDS element distribution; d–Mn–P coating element content

Ref. [20] reported that solid lubricants, such as MoS<sub>2</sub> and graphite, can function as micro balls in the process of friction. According to the surface and interface effect analysis of the material, MoS<sub>2</sub> possesses a large number of dangling keys, exhibiting high chemical instability and reactivity; under the joint action of positive pressure and friction, MoS<sub>2</sub> is easily adsorbed on the friction surface to form a solid lubricating film. Molybdenum disulfide is easily oxidized during the wear process [21], but can prevent the surface adhesion of composite materials and metal materials, thereby effectively improving the wear resistance of the material. The layer structure of graphite reduces the frictional resistance of the friction pair and can effectively reduce friction coefficient and wear amount. Li and Wang reported that MoS<sub>2</sub> exerts a synergistic effect on reducing friction [22, 23]. Furthermore, graphite/MoS<sub>2</sub> composite spraying hard ceramic particles on the substrate enanle to obtain the surface hardening layer. The mechanism of ceramic balls is similar to that of particle shot peening, which results in dense surface to obtain an improved wear resistance [24].

The antifriction mechanism of the manganese phosphate conversion coating is mainly manifested in the

fine adsorption of phosphate film micropores in the Fig. 11 b. The capacity to store lubricating oil improves the lubrication characteristics of the friction surface; after running-in, the "bump" rough peaks of the friction surface are quickly smoothed, the surface becomes smooth, and surface roughness is reduced. In the wear phase of the manganese phosphate conversion coating, the distribution of the coating alloy elements on the grinding surface is illustrated by the EDS of the B-points on the worn surfaces (Fig. 12 b and d). The contents of phosphorus and manganese are very high in the pores; the coating is not worn, and the distribution of the "point" of the phosphate film on the worn surface functionsas oil storage and lubrication to prevent direct contact between the metal surface and adhesion wear [25]. The increase in phosphorus content can gradually smooth the friction surface because of the change in the material microstructure structure caused by phosphorus (appearance of martensite) and the effect on the toughness and hardness of iron-based materials; thus, the shearing properties and adhesion resistance of the matrix material during friction action are improved. Similarly, M. Hamiuddin and D. Ernens concluded that phosphorus can significantly

improve the wear-resistance of iron-based materials [26, 27].

## **4.CONCLUSIONS**

- 1. After surface modification of PC/MoS<sub>2</sub> coating, a solid self-lubricant adheres on the surface of the base material, a solid lubricant adheres on the substrate surface, with a relatively loose coating structure. The solid lubrication medium forms under a high load; in the substrate material interface from the lubricant material, Mo, S, C, and other elements spread to the substrate surface at 20  $\mu$ m depth, strengthening the matrix. The transformation coating of manganese phosphate is irregularly scaled, the coating is denser, and the grain size starts from 0.5  $\mu$ m. The surface contains a small number of pores, and the matrix interface is enriched with P, Mn, and other elements, which can improve the formation of the lubricating oil film.
- In the early run-in stage, PC/MoS<sub>2</sub> and manganese 2. phosphate coating exert an evident antifriction; in the constant grinding phase, the effect of PC/MoS<sub>2</sub> is more prominent. Manganese phosphate conversion coating exhibits a flake porous structure, which can store lubricating oil and is good for oil film formation. The immersion lubrication exhibits better properties of antifriction and wear resistance. By employing solid lubricants  $MoS_2$ , C, and other lubrication  $PC/MoS_2$  reduces the characteristics, friction coefficient of contact pairs. In the aspect of extreme pressure characteristics, PC/MoS<sub>2</sub> exhibits a high extreme pressure antiwear property, and the extreme pressure characteristics are slightly improved by 60% compared with the unprocessed surface under the same experimental conditions. The tribological performance and interfacial mechanism of two different surface modifications will provide a basis for further research on the surface modification of gears and bearings.
- 3. Comparison of the two kinds of surface modification for preparation indicates that the manganese phosphate coating requires simple preparation procedure, short processing period, and low cost, which are suitable for large-scale application. By contrast, PC/MoS<sub>2</sub> spraying features relatively complex equipment and process, high cost, and higher requirements of processing specimens.

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