Sol-gel Derived Europium Doped CaMoO$_4$-Eu$^{3+}$ with Complex Microstructural and Optical Properties

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crossref http://dx.doi.org/10.5755/j01.ms.20.1.4797

Received 16 July 2013; accepted 13 October 2013

The crystalline compouds CaMoO$_4$ doped with Eu$^{3+}$ ions were prepared from nitrate-tartrate precursor gels at 650, 700, 750 and 800 °C temperatures. The obtained samples were characterized by thermal analysis (TG/DSC), Furier Transform infrared spectroscopy (FT-IR), X-ray diffraction (XRD) analysis, scanning electron microscopy (SEM) and reflection measurements (UV-vis). TG/DSC analysis revealed the possible decomposition mechanism of synthesized Ca-Mo-O nitrate-tartrate gel. XRD data exhibited that even at 650 °C temperature the crystalline powelite (CaMoO$_4$) structure has formed. To understand the crystal growth process of the CaMoO$_4$-xEu$^{3+}$ compounds (x = 1, 2, 3, 4, 5 and 6 mol%), the influence of temperature on the surface morphology of the end products by SEM analysis was investigated. No evidence of organic residual in the final materials was found, confirmed by FT-IR analysis. UV-vis reflectance spectra revealed that by increasing of Eu$^{3+}$ ions concentrations in the host material the reflection peaks of Eu$_2$O$_3$ also increased.

Keywords: inorganic compounds, optical materials, sol-gel synthesis, microstructure.

1. INTRODUCTION

In recent years, the optical properties of trivalent rare-earth ions (RE$^{3+}$) in tungstate and molybdate materials with scheelite (CaWO$_4$) and isostructural powelite (CaMoO$_4$) structures have been widely investigated [1]. Especially, many researchers focused their attention on europium (III) ions as luminescence centres in red light phosphors with excellent luminescent properties [2]. It is well known that rare-earth elements have been widely used in high-performance luminescence devices, catalysts and other functional materials because of the electronic, optical and chemical characteristics originating from their 4f electrons [3]. The red luminescence of Eu$^{3+}$ ion has been extensively used in the lighting and displays for its distinct 4f-4f transitions. The f-electron of Eu$^{3+}$ ions are well shielded from the chemical environment and own almost retained atomic character [4].

Metal molybdates are important family of inorganic materials that have great potential applications in various fields, such as phosphors [5], optical fibers [6], scintillators [7], magnets [8] and catalysts [9]. As was mentioned above, calcium molybdate (CaMoO$_4$) crystal possesses the powelite structure (tetragonal, space group I41/a) [10], and has been attracting much attention because of its high performance luminescence devices, catalysts and other functional materials because of the electronic, optical and chemical characteristics originating from their 4f electrons [3]. The red luminescence of Eu$^{3+}$ ion has been extensively used in the lighting and displays for its distinct 4f-4f transitions. The f-electron of Eu$^{3+}$ ions are well shielded from the chemical environment and own almost retained atomic character [4].

Several methods have been developed to prepare CaMoO$_4$ such as traditional solid-state reaction [14], the coprecipitation [15] synthesis, the combustion method [16], Czochralski method [17], solvothermal process [18], reverse micellar reaction [19], microwave irradiation [20], spray pyrolysis [21], the facile microemulsion-mediated hydrothermal process [22], electrochemical method [23], sonochemical preparation [24] and sol-gel synthesis [25]. Among these different synthesis routes the solution based synthetic methods play a crucial role in the design and production of fine ceramics and have been successful in overcoming many of the limitation of the traditional solid-state, high-temperature methods. The use of solution chemistry can eliminate major problems such as long diffusion paths, impurities and agglomeration, which will result in products with improved homogeneity [26]. The metal complexes with organic ligands have been used for the preparation of ceramics and metal oxides thin films by sol-gel process, using metal salts like nitrates [27], chlorides [28], and acetates [29] as starting materials. In this paper we report the synthesis of powellite type CaMoO$_4$ doped with Eu$^{3+}$ ions by a novel aqueous sol-gel synthesis route. In the sol-gel process tartaric acid as a complexing agent has been used.

2. EXPERIMENTAL DETAILS

2.1. Sample preparation

The samples CaMoO$_4$:Eu$^{3+}$ with 1, 2, 3, 4, 5 and 6 mol% of Eu$^{3+}$ were prepared by an aqueous nitrate-tartrate sol-gel route. The europium oxide (Eu$_2$O$_3$, 99.99 %), molybdenum oxide (MoO$_3$, 99.95 %), calcium nitrate tetrahydrate (Ca(NO$_3$)$_2$·4H$_2$O, 99 %) were used as starting materials and weighed according to the desired stiochiometric ratio. Nitric acid (HNO$_3$) and ammonia (NH$_3$·4H$_2$O) were used as solvents and reagents to regulate the pH of the solutions. Tartaric acid (TA, Ca$_2$H$_4$O$_6$, 99.5 %) was used as a complexing agent. In the sol-gel process, MoO$_3$ was dissolved in 25 mL of concentrated ammonia solution by stirring at 70 °C–80 °C. Secondly, tartaric acid with a molar ratio of Mo/TA = 0.25, dissolved in a small amount of distilled water was added with a continuous stirring at the same temperature. Next, after several hours the stoichiometric amount of calcium

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nitrate tetrahydrate dissolved in distilled water was mixed with the previous solution. To prevent precipitation, the excess of ammonia was neutralized with concentrated HNO₃ until the pH reached the value of ~1.0. Afterwards, Eu₂O₃ was added directly to the reaction mixture. Finally, the same amount of the aqueous solution of the complexing agent TA was repeatedly added to the reaction mixture to prevent crystallization of metal salts during the gelation process. The beaker with the solution was closed with a watch glass and left for 1 h with continuous stirring. The obtained clear solution was concentrated by slow evaporation at 80°C in an open beaker. A yellow transparent gel formed after nearly 95 % of the water has been evaporated under continuous stirring. After drying in an oven at 110°C, fine-grained powders were obtained. The precursor gels were calcined for 5 h at 500°C in alumina crucibles and reground carefully in an agate mortar. The five hours of the annealing temperature was chosen because of the full combustion of organic parts in the sample. Since the gels are very combustible, slow heating (1 °C/min), especially between 150°C and 300°C, was found to be essential. After intermediate grinding the obtained powders for 5 h at 650, 700, 750 and 800°C temperatures in air at ambient pressure were annealed.

2.2. Characterization

The thermal decomposition of complex precursor gel was examined by thermogravimetry-differential scanning calorimetry (TG-DSC, Netzsch STA 449C Jupiter instrument) using a sample weight of about 10 mg and a heating rate of 5 °C·min⁻¹ in flowing air (70 cm³·min⁻¹) at ambient pressure from room temperature to 1000°C. The synthesized samples were characterized by X-ray powder analysis (D8 Bruker AXS powder diffractometer) using CuKα radiation. The spectra were recorded at the standard rate of 1.5 2θ/ min. After pressing the samples into the pellets with KBr (~1.5 %), the Fourier transform infrared (FT-IR) spectra were recorded with a Perkin-Elmer FT-IR Spectrum 1000 spectrometer. The scanning electron microscope (SEM) DSN 962 was used to study the surface morphology and microstructure of the obtained ceramic samples. The reflection spectra were recorded at room temperature using a Perkin Elmer Lambda 35 UV/VIS spectrometer. The samples were well glued up on the flat substrate in order to form thin layer on it and excited in 1100 nm – 200 nm wavelength interval.

3. RESULTS AND DISCUSSION

3.1. Thermal analysis

The TG-DSC curves of the CaMoO₄ precursor gel displayed in Fig. 1. These results let us understand its pyrolysis behaviour and crystallization process. The TG curve indicates that by increasing the temperature from up to 530°C the weight loss of the Ca-Mo-O nitrate-tartrate precursor gel occurs. The decomposition process can be roughly divided into three intervals. The first weight loss observed between room temperature and 180°C in the TG curve could be attributed to the removal water from the coordination sphere of the metal complexes and/or of surface adsorbed water. A strong endothermic peak near 175°C in the DSC curve confirms the dehydratation process in this temperature region. With further increasing temperature from 200°C up to 450°C the second main weight loss occurred which is attributed to the pyrolysis of the organic part of the gel and the decomposition of corresponding metal nitrates, as indicated by weight loss about 60 % on the TG curve.

Fig. 1. Combined TG-DSC curves of the CaMoO₄ precursor gel in flowing air

Thereafter, the weight loss of about 10 % observed in the temperature range from 480°C to 520°C is associated with further decomposition of the intermediate pyrolysis products (carbonates or oxy-carbonates). Finally, the weight remains almost constant up to 1000°C, which indicates that the decomposition and combustion of all organic components in the precursor gel is completed below 530°C. The sharp exothermic peak at about 500°C on the DSC curve could be attributed to the final burning of the residual organic species in the Ca-Mo-O gel. The negligible peaks on the DSC curve at ~530°C and ~670°C correspond probably to the crystallization of CaMoO₄. Therefore, we can conclude from Fig. 1, that crystalline CaMoO₄ homogeneously doped with Eu₂O₃ could be obtained at around 650°C, which was proved by XRD measurements.

3.2. X-ray diffraction

Fig. 2 exhibits the X-ray diffraction (XRD) patterns of the CaMoO₄:1 % Eu³⁺ powders obtained by sintering the dried gels for 5 h at 650, 700, 750 and 800°C temperatures. As seen, the calcination of Ca(Eu)-Mo-O nitrate-tartrate precursor gel powders at different temperatures produces a fully crystalline CaMoO₄ phase and no characteristic peaks due to the side crystalline Eu₂O₃ or Eu₂MoO₇ phases were observed. When the dried gels were calcined at 650, 700, 750 and 800°C temperatures only the tetragonal CaMoO₄ phase appeared, which corresponds to ICSD file number 62219. These results clearly indicate that the crystallization of the CaMoO₄ phase is entirely complete at the relatively low temperature of 650°C and this is in a good agreement with the TG-DSC results. However, some low intense impurity peaks in the XRD pattern of the powders calcined at 700°C (see Figs. 2 and 3) could be indexed as a MoO₃ crystalline phase. In addition, no characteristic peaks attributable to the CaO crystal phase were indentified. On the other hand, by further increasing sintering time and temperature the diffraction lines from impurity phase disappeared, and the XRD patterns of the samples annealed at 750°C and 800°C temperatures showed only monophasic crystalline CaMoO₄ (Fig. 2).
It is interesting to note, that this aqueous sol-gel synthesis route from metal nitrate precursors in the sol-gel process using tartaric acid, as a complexing agent could be successful used for the preparation of thin film on glass or silicon substrates. The obtained specimens now are under investigation.

3.3. FT-IR analysis

The infrared spectra of the synthesized CaMoO₄:xEu³⁺ samples annealed at 650 °C for 5 h are shown in Fig. 4. All FT-IR spectra of the CaMoO₄:xEu³⁺ specimens qualitatively are very similar regardless the concentration of europium in the material. Interestingly, the FT-IR spectra of CaMoO₄:xEu³⁺ compounds have approximately the same vibration modes as the CaWO₄:Eu³⁺ phosphor [30].

The bands at 3454 cm⁻¹ and 1631 cm⁻¹ are assigned to O–H stretching vibration and H–O–H bending vibrations [31], respectively. These two bands are the characteristic vibrations of water from air, physically absorbed on the sample surface that is completely different from coordinated water in compound. A strong absorption band at around 910 cm⁻¹ is related to O–Mo–O stretches of the MoO₄ tetrahedron. The AWO₄ type scheelite oxides having S4 symmetry for the WO₄²⁻ groups also show the main absorption bands in the region of 400 cm⁻¹–1000 cm⁻¹, centered around 911, 833 and 405 cm⁻¹ corresponding to the v₁, v₃ and v₂ modes of the WO₄²⁻ groups, respectively [1]. It is also important to note, that the FT-IR spectra of Eu₂O₃ doped calcium molybdate samples calcined at 650 °C did not show any bands attributable to carbonates or residual organic species.

3.4. SEM microscopy

To understand the crystal growth process of the CaMoO₄:xEu³⁺ compounds, we investigated the influence of temperature on the morphology of the products. The SEM micrographs of CaMoO₄:xEu³⁺ samples prepared by the an aqueous sol-gel method at 650, 700, 750 and 800 °C are shown in Figs. 5–10. The SEM results clearly indicate that all products consist of micrometer and sub-micrometer differently shaped particles with narrow crystal size distribution. The homogeneous spherical grains of CaMoO₄:1 % Eu³⁺ with an average diameter of about 1 μm–5 μm have formed when the synthesis was carried out at 650 °C (Fig. 5).

As seen from micrographs at higher magnification (Fig. 5, c and d), the surface morphology of synthesized samples exhibits homogeneous packing of the micrograins where the larger particles (about 5 μm) are surrounded

Fig. 2. XRD patterns of the CaMoO₄:1 % Eu³⁺ precursors heat-treated at 650, 700, 750, 800 °C, and standard ICSD card of tetragonal CaMoO₄ (62219).

Fig. 3. XRD pattern of the CaMoO₄:1 % Eu³⁺ precursors heat-treated at 700 °C and standard ICSD card of MoO₃ (68136).

Fig. 4. FT-IR absorption spectra of CaMoO₄:xEu³⁺ samples.
with smaller grains of about 1 μm diameter. However, the SEM image of CaMoO$_4$: 6 % Eu$^{3+}$ compound (Fig. 6, a) reveals the formation of plate-like larger crystals in size of about 15 μm – 20 μm.

As shown in Fig. 7, a – d, with increasing sintering temperature up to 700 °C two types of grains have formed. The smaller spherical particles, which are less than 200 nm size, tend to aggregate by many small crystal grains. Meanwhile, the larger grains form individual crystals rather than aggregate with each other. This observation could explain the impurity phase of MoO$_3$ in XRD pattern of the CaMoO$_4$: 1 % Eu$^{3+}$ sample prepared at 700 °C for 5 h in air atmosphere.

With further increasing the temperature up to 750 °C, the particles tend to aggregate strongly showing good connectivity between grains (Fig. 8).

Less than 3 μm plate-like crystals with narrow size distribution CaMoO$_4$: 1 % Eu$^{3+}$ sample have formed at this temperature. Fig. 9 shows the SEM micrograph of the CaMoO$_4$: 1 % Eu$^{3+}$ sample synthesized at 800 °C for 5 h in air atmosphere.

However, the surface morphology of the CaMoO$_4$: 1 % Eu$^{3+}$ powders synthesized at 800 °C for 5h are very similar to those prepared at 750 °C. The higher temperature produced just a bit larger crystals. Fig. 10 represents the SEM image of CaMoO$_4$: 6 % Eu$^{3+}$ synthesized at 800 °C for 5 h in air atmosphere. Again, with increasing concentration of europium the larger crystallites of the ceramic material have formed.
The high-magnification SEM images (Fig. 10, c and d) revealed that synthesized products consist of different crystallites with diameter of 2 μm – 4 μm and length 5 μm – 7 μm. These SEM results of the CaMoO$_4$:x % Eu$^{3+}$ samples annealed at different temperatures confirmed that the sintering temperature and the dopant concentration could strongly affect on the morphological features of final products.

3.5. Reflection measurements

The optical properties of doped with europium CaMoO$_4$ ceramics synthesized using the nitrate-tartrate sol-gel technique were also investigated. UV-vis reflectance spectra of CaMoO$_4$:x % Eu$^{3+}$ (x = 1, 2, 3, 4, 5 and 6) phosphor are shown in Figs. 11 – 13.

According to the results presented in [33], CaMoO$_4$ nano-crystals exhibited a very broad asymmetric reflection in the range of 180 nm – 300 nm, which is also observed in Fig. 11 of the CaMoO$_4$:1 % Eu$^{3+}$ samples annealed at 650 °C – 800 °C temperatures. Additionally, the reflectance spectra of all samples are very similar. The reflection bands of CaMoO$_4$:1 % Eu$^{3+}$ samples are located between 200 nm and 315 nm, corresponding to oxygen to molybdenum (O→Mo) ligand-to-metal charge transfer in the MoO$_4^{2-}$ group. As is shown in Fig. 12, the reflection bands at 362, 367, 377, 382, 385 nm and sharp peaks at 394, 465 nm could be attributed to the reflectance of Eu$^{3+}$ ions in the CaMoO$_4$ host-lattice. Moreover, the sharp peak at 394.8 nm tends to increase (see Fig. 13) by increasing dopant concentration in CaMoO$_4$ host-lattice. However, the peak intensity of the CaMoO$_4$:x % Eu$^{3+}$ samples annealed at 750 °C reached the maximum only with doping of 5 mol% of Eu$^{3+}$ ions and further increasing of dopant concentration leads to the quenching of reflectance of obtained optical materials.

As can be seen from Fig. 12, the annealing temperature only little affects the optical properties of CaMoO$_4$ doped with 6 % of Eu$^{3+}$. The maximum intensity of reflection peak at 394.8 nm of excitation light achieved in the sample synthesized at 750 °C temperature.

In conclusion, it should be noted that all synthesized CaMoO$_4$:x Eu$^{3+}$ samples were optically active compounds at shorter wavelengths (200 nm – 310 nm) that arises from MoO$_4^{2-}$. Reflection peaks in 360 nm – 470 nm wavelength region are attributed to the Eu$^{3+}$ characteristic excitations, which are significantly influenced both by annealing temperature and by dopant concentration in CaMoO$_4$ lattice.
4. CONCLUSIONS

Single-phase CaMoO$_4$:x Eu$^{3+}$ (x = 1, 2, 3, 4, 5 and 6 mol% Eu$^{3+}$) phosphors with 1 μm – 7 μm dimensions were prepared by an aqueous nitrate-tartrate sol-gel processing. According to the XRD data, the crystallization of the CaMoO$_4$ precursors is entirely complete at relatively low temperature of 650 °C and is in consonant with the results of TG-DSC. However, by increasing sintering temperature up to 700 °C the small amount of MoO$_3$ impurity phase have formed. With further increasing temperature (750 °C and 800 °C) the XRD patterns of europium doped CaMoO$_4$ showed the formation of only pure crystalline molybdate phase. It was demonstrated, that the morphology and crystallite size of CaMoO$_4$: Eu$^{3+}$ samples are dependent on the annealing temperature and dopant concentration. Moreover, the reflection measurements of the synthesized CaMoO$_4$: Eu$^{3+}$ samples clearly showed that all synthesized CaMoO$_4$: xEu$^{3+}$ specimens are optical active compounds.

Acknowledgments

The study was funded from the European Community’s social foundation under Grant Agreement No. VPI-3.1-ŚMM-08-K-01-004/KS-120000-1756.

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