Effects of Solution Concentration on the Properties of Cu₄SnS₄ Thin Films

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Copper tin sulfide thin films were electrodeposited on the indium tin oxide substrate from an aqueous solution containing $CuSO_4$, $SnCl_2$ and $Na_2S_2O_3$ at pH 1. Deposition at various concentrations was attempted in order to study the effect of electrolytes concentration on the film properties. The thin films were characterized using X-ray diffraction and atomic force microscopy. The absorption properties, band gap energy and transition type was determined using UV-Vis Spectrophotometer. The thin films produced were polycrystalline in nature. The XRD data showed that the most intense peak is at $2\theta = 30.2^{\circ}$ which belongs to (221) plane of Cu_4SnS_4 . The AFM images indicated that the lower concentration leads to smaller crystal size, as well as higher optical absorption values. The optimum bath composition was found to be 0.01 M for $CuSO_4$, $Na_2S_2O_3$ and $SnCl_2$. The band gap value was found to be 1.7 eV with indirect transition. *Keywords*: bandgap energy, electrodeposition, semiconducting material, solar cells, thin films.

INTRODUCTION

The conversion of sunlight directly into electricity using the electronic properties of suitable materials is one of the most elegant energy conversion processes. The solar cell technology has been enormous development during the last three decades, initially for providing electrical power for spacecrafts and more recently for terrestrial applications. The present solar cells industry is dominated by crystalline silicone based solar cells. However, silicon supply is the problem limiting the future of their market. Thin film solar cells, which need merely a very thin semiconductor film have advantages in the lowering the material cost.

There are many methods for preparing thin films such as chemical bath deposition [1], vacuum evaporation [2], electrodeposition [3], molecular beam epitaxy [4], close spaced sublimation [5], thermal evaporation [6], spray pyrolysis [7], sputter deposition [8], metal organic chemical vapor deposition [9] and plasma-enhanced chemical vapor deposition [10]. Amongst these deposition methods, electrodeposition is more attractive, since it offers the advantages of simplicity, economy, convenience and several experimental parameters can be controlled more precisely. To this date, many semiconductor thin films deposited by electrodeposition have been reported including Cu₂S [11], SnS [12], CdSe [13], CdS [14], PbS [15], PbSe [16], SnSe [17], ZnS [18], CdIn₂Se₄ [19], ZnCuTe [20], SnS_{0.5}Se_{0.5} [21], CuInS₂ [22] and CuInSe₂ [23].

In this paper, we report the morphological, structural and optical properties of Cu_4SnS_4 thin films obtained by electrodeposition method at various concentrations of $CuSO_4$, $SnCl_2$ and $Na_2S_2O_3$. The films have been characterized by X-Ray Diffraction for structure determination, atomic force microscope for surface morphology study and UV-Visible Spectrophotometer for optical absorption properties study.

EXPERIMENTAL PROCEDURE

The electrodeposition was carried out in а conventional three-electrode cell. The indium-doped tin oxide (ITO) was used as the working electrode. The counter electrode and reference electrode was made from platinum wire and Ag/AgCl, respectively. The ITO was cleaned ultrasonically in ethanol and distilled water before use. The EG&G Princeton Applied Research potentiostat driven by a software model 270 Electrochemical Analysis System was used to control electrodeposition process and to monitor current and voltage profiles. The solutions used were CuSO₄, SnCl₂, Na₂S₂O₃ and HCl. The experiment was performed at room temperature without stirring. The pH was maintained at 1 using 0.05 M of HCl. HCl was added to prevent the formation of hydroxyl species and insoluble compounds. Also, HCl used to completely dissolve the tin chloride to result in a clear transparent solution. In order to investigate the effect of electrolytes concentration on the film properties, deposition at various concentrations was carried out. The first set of experiment was carried out using constant concentration of 0.01 M of CuSO₄, SnCl₂ and varying concentrations of Na₂S₂O₃ (0.01 M - 0.02 M) solutions. The second set of experiment was carried out using fixed concentration of 0.01 M of CuSO₄, Na₂S₂O₃ and varying concentrations of SnCl₂ (0.01 M - 0.02 M) solutions. The third set of experiment was carried out using constant concentration of 0.01 M of $SnCl_2$, $Na_2S_2O_3$ and varying concentrations of $CuSO_4$ (0.01 M - 0.02 M) solutions. The deposition was carried out at -600 mV vs. Ag/AgCl for 45 minutes. After the deposition, the films were washed with distilled water and kept for analysis.

X-ray diffraction (XRD) analysis was carried out, using a Philips PM 11730 diffractometer for the 2θ ranging from 25° to 60° with CuK_{α} ($\lambda = 1.5418$ Å) radiation. Topography was measured by using an atomic force microscopy (Quesant Instrument Corporation, Q-Scope 250) operating in contact mode, with Si₃N₄ cantilever. The photoelectrochemical experiments were performed in

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 $[Fe(CN)_6]^{3-}/[Fe(CN)_6]^{4-}$ redox system by running linear sweep voltammetry (LSV) between two potentials limits (1.0 V to -1.0 V). The sequence of constant illumination, chopped illumination and dark period were performed on the PEC cell to study the effect on photoactivity behavior. A halogen lamp (300 W, 120 V) was used for illuminating the electrode. Optical absorption study was carried out using the Perkin Elmer UV/Vis Lambda 20 Spectrophotometer. The film-coated indium doped tin oxide (ITO) glass was placed across the sample radiation pathway while the uncoated ITO glass was put across the reference path. The absorption data were manipulated for the determination of the band gap energy, E_g .

RESULTS AND DISCUSSIONS

Fig. 1 shows the XRD patterns of the films deposited at various $CuSO_4$ concentrations (0.01 M - 0.02 M) and constant Na₂S₂O₃, SnCl₂ at 0.01 M. There are six Cu₄SnS₄ peaks at $2\theta = 28.6^{\circ}$, 30.1° , 35.2° , 42.9° , 45.2° and 50.6° for the samples prepared at 0.01 M, 0.015 M and 0.02 M of CuSO₄. The corresponding interplanar distances are well in agreement with JCPDS data (Reference code: 010710129) of 3.12, 2.96, 2.54, 2.10, 2.00 and 1.80 Å, which attributed to the (102), (221), (420), (331), (512) and (711) planes, respectively. All these peaks are related to the compound of Cu_4SnS_4 of orthorhombic structure (a = 13.5580 Å, b = 7.6810 Å, c = 6.4120 Å, $\alpha = \beta = \gamma = 90^{\circ}$). However, as the concentration of CuSO₄ was higher than 0.015 M and 0.02 M, the copper sulfide peaks (Reference code: 000653556) which corresponding to 2.82 Å and 2.32 Å at $2\theta = 31.8^{\circ}$ and 38.8° , respectively were obtained.

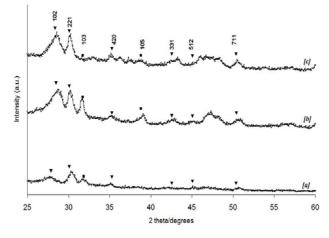
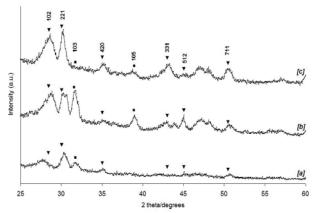
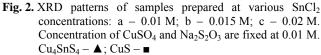


Fig. 1. XRD patterns of samples prepared at various $CuSO_4$ concentrations: a - 0.01 M; b - 0.015 M; c - 0.02 M. Concentration of $SnCl_2$ and $Na_2S_2O_3$ are fixed at 0.01 M. $Cu_4SnS_4 - \blacktriangle$; $CuS - \blacksquare$

Fig. 2 shows the XRD patterns of the films deposited at various SnCl_2 concentrations (0.01 M – 0.02 M) and fixed Na₂S₂O₃, CuSO₄ at 0.01 M. XRD indicates the presence of six peaks at $2\theta = 28.5^{\circ}$, 30.1° , 35.1° , 42.8, 45.2° and 50.5° belonging to Cu₄SnS₄ for samples prepared using lower concentrations (0.01 M and 0.015 M). There is no copper sulfide peaks were observed from the samples deposited with 0.01 M of tin chloride. Six peaks corresponding to interplanar distance of 3.12, 2.96, 2.55, 2.11, 2.01 and 1.80 Å were observed for the film prepared from 0.02 M SnCl_2 .





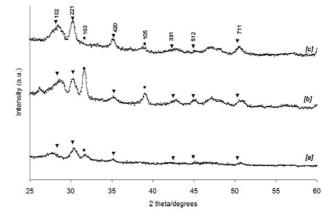


Fig. 3. XRD patterns of samples prepared at various $Na_2S_2O_3$ concentrations: a - 0.01 M; b - 0.015 M; c - 0.02 M. Concentration of $CuSO_4$ and $SnCl_2$ are fixed at 0.01 M. $Cu_4SnS_4 - \blacktriangle$; $CuS - \blacksquare$

Fig. 3 shows the XRD patterns of the films deposited at various Na₂S₂O₃ concentrations (0.01 M – 0.02 M) with constant CuSO₄, SnCl₂ at 0.01 M. The thin films prepared in different concentrations of Na₂S₂O₃ showed six peaks at $2\theta = 28.9^{\circ}$, 30.1°, 35.1°, 42.8, 45.2° and 50.7°, corresponding to d-spacing values 3.08, 2.96, 2.55, 2.11, 2.01 and 1.79 Å, which attributed to the (102), (221), (420), (331), (512) and (711) planes, respectively were detected. The appearances of copper sulfide peaks were detected when the concentration of Na₂S₂O₃ was higher at 0.015 M and 0.02 M.

Fig. 4 shows the AFM images of films prepared at different $CuSO_4$ concentrations and constant $SnCl_2$, $Na_2S_2O_3$ at 0.01 M. The grain size for the film prepared at 0.015 M and 0.02 M are almost similar and do not much different from each other (Fig. 4, b, c). The crystal size decreases with the decrease of $CuSO_4$ concentration (Fig. 4, a).

Fig. 5 shows the AFM images of films prepared at different $SnCl_2$ concentrations and constant $Na_2S_2O_3$, $CuSO_4$ at 0.01 M. The images indicated that higher concentration of $SnCl_2$ leads to larger crystal size (Fig. 5, b, c) while lower $SnCl_2$ exhibits smaller crystal size (Fig. 5, a).

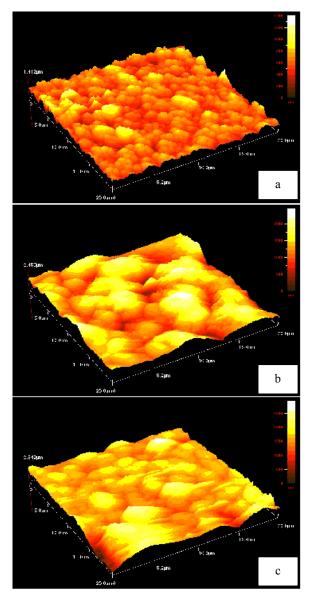


Fig. 4. Atomic force microscopy images of Cu_4SnS_4 films deposited at various $CuSO_4$ concentrations: a - 0.01 M; b - 0.015 M; c - 0.02 M. Concentration of $Na_2S_2O_3$ and $SnCl_2$ are fixed at 0.01 M

Fig. 6 shows the AFM images of films prepared at different $Na_2S_2O_3$ concentrations and constant $SnCl_2$, $CuSO_4$ at 0.01 M. The images pointed out that the deposits are crystalline and their grain size varies with the variation of $Na_2S_2O_3$ concentrations. The higher $Na_2S_2O_3$ concentration leads to bigger crystal size (Fig. 6, b, c) while lower $Na_2S_2O_3$ exhibits smaller crystal size (Fig. 6, a).

Figs. 7–9 shows the absorption spectra of Cu_4SnS_4 films at different concentrations of $CuSO_4$, $SnCl_2$ and $Na_2S_2O_3$, respectively. The films show a gradually increasing absorbance throughout the visible region, which makes it possible for this material to be used in a photoelectrochemical (PEC) cell. From the graph, it is indicated that the samples prepared at lower $CuSO_4$, $SnCl_2$ and $Na_2S_2O_3$ concentration (0.01 M) have higher absorption values respectively. Thus, this concentration is more preferable in the synthesis of Cu_4SnS_4 films of better quality on ITO substrate. The optical absorption values are in line with AFM results, which the larger grain size produces lower absorption results.

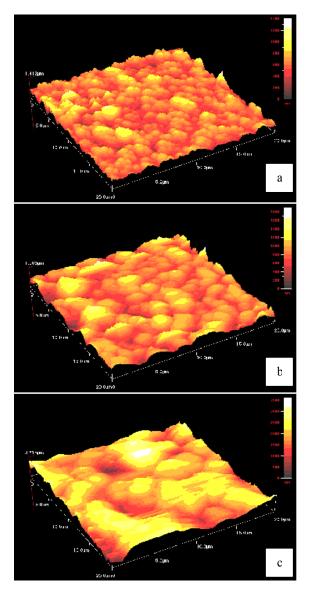


Fig. 5. Atomic force microscopy images of Cu_4SnS_4 films deposited at various $SnCl_2$ concentrations: a - 0.01 M; b - 0.015 M; c - 0.02 M. Concentration of $Na_2S_2O_3$ and $CuSO_4$ are fixed at 0.01 M

Band gap energy and transition type can be derived from mathematical treatment of data obtained from optical absorbance versus wavelength with Stern relationship of near-edge absorption:

$$A = \frac{[k(hv - E_g)^{n/2}]}{hv} ,$$
 (1)

where v is the frequency, h is the Planck's constant, k equals a constant while n carries the value of either 1 or 4. The band gap, E_g , could be obtained from a straight line plot of $(Ahv)^{2/n}$ as a function of hv. Extrapolation of the line to the base line, where the value of $(Ahv)^{2/n}$ is zero, will give E_g . The $(Ahv)^{1/2}$ versus hv plot is a straight line (Fig. 10) indicating that the energy band gap of Cu₄SnS₄ is indirect and intercept on the hv axis yield a band gap of 1.7 eV for the film prepared using 0.01 M Na₂S₂O₃, SnCl₂ and CuSO₄.

Fig. 11 shows the photoresponse of the Cu_4SnS_4 thin films grown from 0.01 M $CuSO_4$, $SnCl_2$ and $Na_2S_2O_3$ in contact with 0.01 M Fe^{2+}/Fe^{3+} redox couple. An increase in

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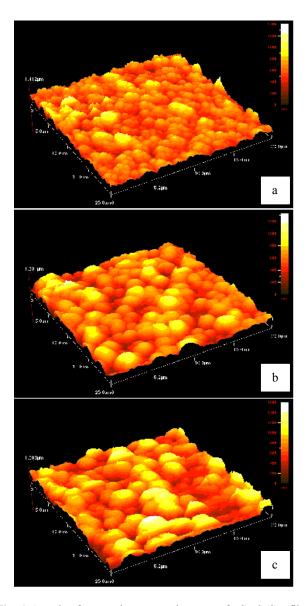


Fig. 6. Atomic force microscopy images of Cu_4SnS_4 films deposited at various $Na_2S_2O_3$ concentrations: a - 0.01 M; b - 0.015 M; c - 0.02 M. Concentration of $SnCl_2$ and $CuSO_4$ are fixed at 0.01 M

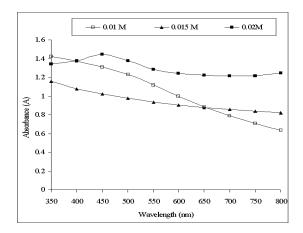


Fig. 7. Optical absorbance versus wavelength of the Cu_4SnS_4 films deposited at various $CuSO_4$ concentrations (0.01 M - 0.02 M). Concentration of $SnCl_2$ and $Na_2S_2O_3$ are fixed at 0.01 M

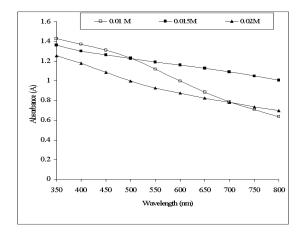


Fig. 8. Optical absorbance versus wavelength of the Cu_4SnS_4 films deposited at various $SnCl_2$ concentrations (0.01 M - 0.02 M). Concentration of $CuSO_4$ and $Na_2S_2O_3$ are fixed at 0.01 M

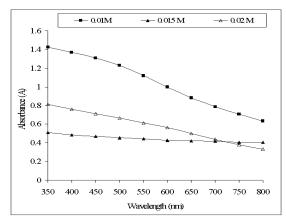


Fig. 9. Optical absorbance versus wavelength of the Cu_4SnS_4 films deposited at various $Na_2S_2O_3$ concentrations (0.01 M – 0.02 M). Concentration of $CuSO_4$ and $SnCl_2$ are fixed at 0.01 M

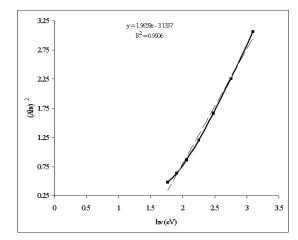


Fig. 10. Plot of $(Ahv)^{2/n}$ versus hv when n = 4 for Cu_4SnS_4 films prepared using 0.01 M Na₂S₂O₃, CuSO₄ and SnCl₂ solutions

current could be observed as the films were illuminated. The current change upon illumination indicates semiconductor behavior of the materials. The fact that photocurrent occurs on the positive potential region reflects that the films prepared are n-type. This indicates that the minority carriers generated are holes.

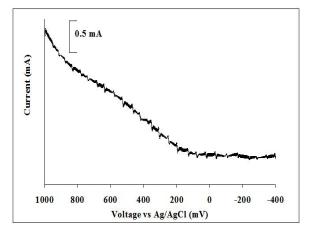


Fig. 11. The photoresponse of Cu_4SnS_4 films prepared using 0.01 M $Na_2S_2O_3$, $CuSO_4$ and $SnCl_2$ solutions

CONCLUSIONS

The Cu₄SnS₄ thin films can be electrodeposited using CuSO₄, SnCl₂ and Na₂S₂O₃ solution. X-ray diffraction data showed that the intensity of major peaks at 2.96 Å that attributed to (221) plane of Cu₄SnS₄. The AFM images indicated that higher CuSO₄, SnCl₂ and Na₂S₂O₃ concentration leads to larger crystal size. The films with smaller crystal size and show better photoresponse have been obtained from lower electrolytes concentration (0.01 M). The optimum bath composition was found to be 0.01 M for CuSO₄, SnCl₂ and Na₂S₂O₃ to produce good quality thin films. The films exhibited n-type semiconductor behavior with band gap energy of 1.7 eV.

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